Article

Solubilities of Ethylene and Carbon Dioxide Gases in Lithium-Ion Battery Electrolyte

Mel Soto, Kae Fink, Christof Zweifel, Peter J. Weddle, Evan Walter Clark Spotte-Smith, Gabriel M. Veith, Kristin A. Persson, Andrew M. Colclasure, and Bertrand J. Tremolet de Villers*



ethylene carbonate/ethyl methyl carbonate and 3 wt % fluoroethylene carbonate (15:25:57:3 wt % total composition). The experimentally measured pressure decay curve is fit to an analytical dissolution model and extrapolated to predict the final pressure at equilibrium. The relationship between the partial pressures and concentration of dissolved gas in electrolyte at equilibrium is then used to determine Henry's law constants of $k_{C_2H_4} = 2.0 \times 10^4$ kPa for C_2H_4 and $k_{CO_2} = 1.1 \times 10^4$ kPa for CO_2 . These values are compared to Henry's law constants predicted from density functional theory and show good agreement within a factor of 3.

INTRODUCTION

Li-ion batteries are currently one of the most energy-dense commercial battery chemistries, dominating the market for electronic devices, electric automotives, and stationary energy storage.¹⁻³ However, further energy density improvements are required to electrify markets such as flight, freight, and maritime transport.¹⁻⁴ In pursuit of next-generation batteries with even higher energy densities, new chemistries for anodes, cathodes, and electrolytes are continuously being investigated. At present, some of the materials receiving the most interest and scrutiny include silicon or lithium metal as replacements for graphite anodes,⁵⁻⁹ lithium-, manganese-, and nickel-rich cathodes,^{10,11} and ether-based localized high-concentration electrolytes.¹² However, some of these materials have yet to achieve mass commercialization due to significant shortcomings in cycle and/or calendar lifetimes as a result of unstable reactivities between the electrodes and the electrolyte.

A stable, electronically passivating, and ion-permeable solidelectrolyte interphase (SEI) is a key feature of Li-ion batteries, as it protects the electrode from continuous side reactions with the electrolyte while enabling Li-ion transport. While various additives, usually liquids, have been explored to improve the stability and permeability of anode SEIs, relatively little research has been done to determine the role of gases in the performance of commercial Li-ion battery chemistries, let alone next-generation battery compositions.^{5,13} Gas generation and subsequent consumption at the anode SEI has been documented in nondegassed pouch cell systems.¹⁴ Many gases are generated during Li-ion battery cycling, any of which could have possible beneficial or adverse effects on performance. Early work on Li-metal anode and Li-graphite anode batteries explored CO₂ as an additive, finding it improved cycling efficiency $^{15-18}$ and stability of the graphite SEI. ^{15,19,20} These beneficial consumptive effects of CO₂ are typically associated with a more favorable SEI through the suppression of transesterification reactions and production of more crosslinked poly(ethylene oxide)-type polymeric species.^{17,21} More recently, Blaubaum et al. studied Li-ion batteries containing electrolyte saturated with CO2, CO, C2H4, C2H2, H2, CH4, and O₂ (all gases that are commonly reported in Li-ion batteries), and concluded that battery electrolytes saturated with CO2 and O₂ showed higher C-rate capabilities and less irreversible capacity loss during the first cycle.¹³

Received:November 21, 2023Revised:May 9, 2024Accepted:May 10, 2024

Downloaded via LAWRENCE BERKELEY NATL LABORATORY on May 23, 2024 at 18:22:06 (UTC) See https://pubs.acs.org/sharingguidelines for options on how to legitimately share published articles.



pubs.acs.org/jced

Tabl	e 1	. (Chemicals	Used	in	this	Stud	y, 8	Supp	liers,	Purity,	and	Purif	ication	Metho	ds
------	-----	-----	-----------	------	----	------	------	------	------	--------	---------	-----	-------	---------	-------	----

chemical	supplier	initial purity, mol fraction	purification method	final purity, mol fraction	analysis method
15 wt % lithium hexafluorphosphate (LiPF ₆), 25.5 wt % EC, 59.5 wt % EMC ^a	Tomiyama Pure Chemical Ind., Ltd.	0.9998		0.9998	
FEC	Tomiyama Pure Chemical Ind., Ltd.	0.99		0.99	
carbon dioxide (CO ₂)	GASCO	0.9999		0.9999	
ethylene (C ₂ H ₄)	Sigma-Aldrich	≥0.995		≥0.995	
water	Tap		Barnstead E-pure ultrapure water purification system	>0.9999	resistivity

^aFormulation was purchased premixed by the supplier, and the uncertainty of the composition is unknown.

Gas-phase byproducts typically result from electrochemical oxidation/reduction of the electrolyte solvent species. In the electrolyte system evaluated in this work, fluoroethylene carbonate (FEC) has the highest reduction potential, and thus is most likely to decompose to form species such as vinylene carbonate and lithium fluoride (LiF).^{22,23} Ideally, gas species formation could be used as a signature of a particular electrolyte species reduction, if the reaction mechanisms for electrolyte decomposition are known.²⁴ For example, CO is typically attributed as a byproduct from the reduction of ethyl methyl carbonate (EMC) and subsequent transesterification to diethyl carbonate and dimethyl carbonate (DMC).^{23,25,26} As mentioned above, the presence of CO_2 is generally reported to have favorable effects on the cycle life.17,18,27 However, competing pathways are sufficiently complex that attributing a particular gas-phase product to a certain liquid-phase reactant has proven difficult. When trying to untangle the influence of a particular species on reaction pathways, it is important to consider whether dissolved species are preferentially retained in the electrolyte or expelled away from the electrode into the reduction-free gas phase. For example, if the thermodynamics of CO₂ enable greater stability as a gas-phase species compared to a dissolved liquid-phase species, reaction mechanisms involving CO₂ consumption in the bulk, liquid electrolyte would be less likely.

To understand the availability of a particular gaseous species at the electrode interface to form the SEI, the gas species solubility in the electrolyte must be known. The solubility of nonpolar gases, including ethylene, in DMC has been studied previously by a gas saturation method measuring volume of gas dissolved into a known volume of solvent.^{28,29} Ethylene solubility in other common battery solvents such as ethylene carbonate (EC) and EMC has not yet been reported. Further, previous studies on CO₂ solubility in common electrolyte solvent mixtures employed a method involving full gas saturation and subsequent displacement and chemical titration of the dissolved CO2. 30,31 These studies indicate that the addition of Li salts such as LiPF₆ can also increase the solubility of the gas, emphasizing the need to study real electrolyte systems. Each of these two methods were developed to study a specific subset of gas-phase species, but each also has notable limitations. In particular, the latter approach (chemical titration), which was used to study CO_2 is poorly suited to detect nonpolar gases; the former method (measuring the volume of dissolved gas in solvent) is better-suited for nonpolar gases, but introduces temperature differences over the entire system that require correction factors. Thus, neither of these approaches is optimal to conduct solubility measurements for the practical application of battery electrolyte interfaces, where precise quantification of solubility across a

broad class of gaseous species is required. Instead, a method for investigating solubility of both polar and nonpolar gases with a temperature-controlled closed system pressure differential technique is employed in this work. Solubility values are obtained by monitoring the pressure decay of a gas above a liquid at a constant temperature until a steady state is reached.^{32,33} The measured change in pressure corresponds to the concentration of gas in the saturated liquid at the final steady state.

The pressure decay technique often requires long time scales to reach equilibrium, especially in systems with high viscosity where the measurements take days to weeks.^{32,33} To improve experimental measurement throughput, the final equilibrium pressure $P_{\rm eq}$ can be extrapolated from initial data using a nonlinear regression based on the molar flux of diffusion outlined in Fick's laws. To this end, an equation derived by Behzadfar and Hatzikiriakos³² has been used in this study to model the pressure decay of the system given initial pressure decay data to return $P_{\rm eq}$. Discussion of assumptions used in this model and the subsequent calculation of solubility is included in the pressure decay modeling section below.

We present an apparatus and associated methodology to determine the solubility of carbon dioxide and ethylene gases in a battery electrolyte by measuring the pressure change during dissolution of the gases into the liquid. The gaseous species concentration in the liquid has been plotted against the pressure of the gas above the liquid to determine the Henry's law constant, k

$$P_{\rm eq} = k \cdot \chi \tag{1}$$

where χ is the saturated gas solubility as a mole fraction and k is the Henry's law constant in kPa.

EXPERIMENTAL SECTION

Chemicals used in this study, their suppliers, and other relevant information are summarized in Table 1.

Pressure Decay Trials. The pressure decay experiments were conducted in a custom-built stainless-steel Swagelok cell, modified from a design developed by Oak Ridge National Laboratory (Figure S1).³⁴ The pressure was monitored using an Omega PX-409 USBH pressure transducer. At the start of each pressure-decay trial, the entire cell was evacuated to approximately 5.9 kPa, and then filled with the target gas to the desired pressure. In hard-cased silicon-based Li-ion batteries, internal pressures have been reported to vary from approximately 300–950 mbar relative to atmospheric pressure due to expansion and contraction during cycling.³⁵ Considering a standard atmospheric pressure of 101 kPa, the absolute pressures of gas in the cell ranges from 131 to 196 kPa. Therefore, three filling pressures were chosen: 18, 28, and 40

psia, corresponding to target gas initial partial pressures ranging from 124 to 276 kPa. After gas filling, a needle valve was closed to trap gas in the gas reservoir. The other side of the cell was opened to allow the remaining gas to escape and remained open as the apparatus was cycled through an antechamber into a glovebox with argon atmosphere.

Inside the glovebox, 2 mL of electrolyte (comprising 15:25:57:3 wt % LiPF₆/EC/EMC/FEC, hereafter referred as GenF3) was added into the electrolyte reservoir using a syringe with an 18 gauge, 8" needle. The argon pressure in the box was recorded and a ball valve was closed to seal the cell. The cell was brought out of the glovebox and placed into a temperature-controlled chamber at 30 °C (303 K). The cell was allowed to rest for 2 h to allow the internal pressure to equilibrate. The equilibrated value was recorded as $P_{\rm gr}$ and indicates the pressure of the gas reservoir before the trial began. The needle valve was then fully opened to release the target gas into the electrolyte chamber. The pressure decay over time was recorded in 3 s intervals on the Omega Transducer software.

Accurate volume measurements of the entire cell were needed to calculate the moles of gas present in the system during the trials. This was done separately using a differencein-mass method with deionized water. During volume measurements, the transducer was replaced by a solid steel NPT plug, which was tightened to the same depth as the transducer. The cell with all valves open was fully dried at 80 °C (353 K) overnight in a vacuum oven, then cooled and weighed empty. The cell was then filled with water through the electrolyte reservoir, capped, and then reweighed. This was repeated and the average weight was converted to volume using the density of water. The volumes for the gas and electrolyte chambers were determined by experiment. Briefly, the gas reservoir was filled with air, the needle valve was closed, and the electrolyte chamber was evacuated to approximately 5.9 kPa. The ball valve was closed to isolate the entire cell and the cell was placed in an oven to equilibrate at 30 °C. Following equilibration and recording of the gas reservoir initial pressure, P_{ov} , the needle valve was opened to allow the gas on the gas reservoir side to expand and fill the vacuum. This final pressure at equilibrium was recorded. The ratio of the pressures before and after the release of the gas were used to calculate the volume ratio of the gas reservoir side to the entire cell volume.

To evaluate the capability of the measurement apparatus and test the validity of our data analysis methods (described in more detail in the next section), the Henry's law constant of carbon dioxide in water at 30 °C (303 K) was measured, Figures S2 and S3. The experimental solubility value was found to be 1.92×10^5 kPa; data is provided in Table S3. Carbon dioxide solubility in water has been extensively characterized with a reported Henry's law constant of 1.85×10^5 kPa at 303.15 K,³⁶ suggesting that the present experimental method has a high degree of accuracy for prediction of Henry's law constant values, Table S4.

Pressure Decay Modeling. The pressure decay of a gas dissolving into a liquid through mass transfer was modeled by Behzadfar and Hatzikiriakos³² and results in the analytical expression given by

$$P = P_{eq} + \frac{8c_{sat}Z_g RTV_1}{\pi^2 V_g} \sum_{n=1}^{\infty} \frac{1}{(2n-1)^2} \exp\left(-(2n-1)^2 \frac{\pi^2}{4L^2} Dt\right)$$
(2)

where P is the measured pressure, c_{sat} is the saturation concentration in mole fraction, Z_g is the gas compressibility (0.99426 and 0.99 for C_2H_4 and CO_2 , respectively), V_1 is the volume of the liquid (2.0 mL), D is the species liquid-phase diffusivity, L is the diffusion length (7.02 mm), R is the universal gas constant, T is the temperature, and V_{σ} is the volume of the gas (12.78 mL), see also Table S1. Equation 2 solves for the pressure at time t, and notably does not model the initial pressure drop due to the interface filling phenomenon that occurs at the beginning of the experiment.³² To use this expression, our data sets exclude this early stage pressure drop to obtain a more accurate regression. It is assumed that the electrolyte, prepared only in the glovebox, has little to no C_2H_4 or CO_2 gas dissolved it in to begin with. The data sets are fit to eq 2 by iteratively solving for the final pressure, P_{eq} , and the diffusivity D that minimize the sum of squares of the error differences between the experimental and model data points (i.e., least-squares regression fitting).

After all trials at the same gas reservoir pressure had concluded and been modeled, their diffusivities were averaged and input back into eq 2. The fit was then determined by keeping D constant and only varying $P_{\rm eq}$. These final produced $P_{\rm eq}$ values were used for subsequent solubility calculations and the construction of the Henry's law curve.

It is important to note that $c_{\rm sat}$ in the Behzadfar model is calculated from the first pressure point of the data used for regression and the solved variable $P_{\rm eq}$.³² As this necessarily excludes the early stage pressure drop and involves the total pressure, the real $c_{\rm sat}$ value was calculated separately for the Henry's law graphs. The solubilities were calculated from the difference in target gas moles between the initial and equilibrium conditions. The target gas initial and equilibrium partial pressures were calculated by subtracting the partial pressure of argon from the glovebox above the electrolyte and the vapor pressure of a similar electrolyte at 25 °C (1 M LiPF₆ in 3:7 wt/wt EC/EMC)³⁷ from each trial's $P_{\rm eq}$ value. These Ar and vapor pressures were assumed constant throughout the duration of each trial (Table S2).

To supplement and corroborate experimental measurements, the Henry's Law constant was also predicted using density functional theory (DFT). All DFT calculations used the Q-Chem electronic structure code version 5^{38} with the wB97X-V range-separated hybrid generalized gradient approximation exchange-correlation functional³⁹ and the def2-TZVPPD basis set.⁴⁰ Optimized structures, electronic energies (E), enthalpies (H), and entropies (S) corresponding to CO_2 and C_2H_4 in solution were taken from the lithium ion battery electrolyte (LIBE) data set.²⁶ In LIBE, solvent effects are treated implicitly using the solvent model with density (SMD),⁴¹ using parameters relevant to a 3:7 wt/wt mixture of EC and EMC. SMD is based on the polarizable continuum model,⁴² which models a bulk solvent environment as a uniform dielectric medium surrounding a solute-shaped cavity. In addition to the PCM-like bulk electrostatic terms, SMD adds energy terms accounting for short-range interactions, namely cavitation, dispersion, and local solvent structure. The solvent-optimized CO₂ and C₂H₄ structures from LIBE were reoptimized in vacuum, and then the vacuum-optimized structures were subjected to a vibrational frequency analysis to obtain the gas-phase thermochemistry (e.g., H, S).

Molecular free energies are calculated from DFT as

$$G = H - TS \tag{3}$$

where the enthalpy term (*H*) implicitly includes the electronic energy and zero-point energy. For this study, we consistently used $T = 303.15 \text{ K} = 30 \text{ }^{\circ}\text{C}$, as our experimental measurements were taken at 30 $^{\circ}\text{C}$.

The solvation free energy $\Delta G_{\text{solvation}}$ is calculated as

$$\Delta G_{\rm solvation} = G_{\rm EC/EMC} - G_{\rm vacuum} \tag{4}$$

where $\Delta G_{\rm EC/EMC}$ is the free energy of the molecule in the solution phase (using SMD) and $\Delta G_{\rm vacuum}$ is the free energy of the molecule in vacuum. From $\Delta G_{\rm solvation}$, the Henry's law coefficient k can be expressed^{43,44} as

$$k = RT \exp(\Delta G_{\text{solvation}} / RT)$$
(5)

DFT calculations using SMD can predict the solvation free energies of small molecules with high accuracy. When calculating the solvation free energies of neutral molecules in one of the 90 nonaqueous solvents included in its training set, SMD achieves a mean unsigned error of 0.67 kcal mol^{-1} (0.03 eV).⁴¹ While one might reasonably expect a somewhat higher error when calculating solvation free energies in solvents outside of the training set, such as 3:7 wt/wt EC/EMC, the thermodynamics obtained from DFT should nonetheless be reasonably accurate for the types of small gases considered here. However, calculating Henry's law coefficients is considerably more challenging as compared to computing the solvation free energies. As seen in eq 5, calculating the Henry's law coefficient requires exponentiation of the solvation free energy, meaning even very small errors in $\Delta G_{\text{solvation}}$ can have a considerable impact on the predicted coefficient. As seen in eq 5, modifying $\Delta G_{\text{solvation}}$ by 1 kcal mol⁻¹ would cause k to change by a factor of roughly 5.4.

RESULTS

Figure 1 illustrates a long-term trial of C_2H_4 dissolution in GenF3 with a gas reservoir initial pressure, $P_{gr} = 288.4$ kPa.



Figure 1. A pressure decay curve of C_2H_4 dissolution into GenF3 with a starting gas reservoir pressure, $P_{gr} = 288.4$ kPa. Equilibrium of the C_2H_4 gas pressure above the liquid and the saturated concentration of C_2H_4 in the liquid was reached after 1060 h at T = 303.15 K.

After the needle valve is opened, the gas escapes the reservoir and fills the entire cell, which causes a sharp pressure drop due to expansion of the gas into the part of the cell containing electrolyte (the headspace above the liquid). This is followed by a steep initial pressure decay, attributed to the interfacefilling phenomenon as the gas saturates the topmost layers of the liquid in the reservoir.³² The final value P_{eq} is an average of the last 20 measured data points before the trial was stopped after 1060 h. The initial and equilibrium partial pressures of the gas are used to accurately calculate the change in moles moving from the gas phase to the solution phase.

Equation 2 was also used to extrapolate the P_{eq} of shorter trials (<150 h) that were concluded before equilibrium was reached. To best approximate the behavior of the slow pressure decay due to dissolution, the data sets used in the model are in seconds and begin where $dP/dt < 0.034 \Delta kPa s^{-1}$, after the interface-filling regime. To determine an optimal time for the conclusion of the trials, the t = 1060 h data set was used to calculate the model-fit percent error in P_{eq} when the trial runtime length was varied. Figure 2 shows the decreasing percent error with data sets of increasing trial runtime. After 96 h the percent error is consistently less than 1%. There is some systematic noise in Figure 2 that appear as "bumps" in the data appearing approximately 24 h apart. This systematic error is attributed to daily temperature changes in the room.

Figure 3 shows the experimental and best-fit curves for a C_2H_4 and a CO_2 trial at $P_{gr} = 193$ kPa. The fits have a coefficient of determination, $R^2 > 0.99$, for both gases. Although the Behzadfar and Hatzikiriakos model was developed to evaluate the diffusivity of CO_2 in bitumen,³² the observed $R^2 \approx 1$ for both gases studied here, despite large differences in decay rates, emphasizes the accuracy of the chosen model for a carbonate-based battery electrolyte.

Figure 4 illustrates the solubility with respect to partial pressure for (a) C_2H_4 and (b) CO_2 . The data for the figure and their uncertainties are listed in Tables 2 and 3. The slope of the lines is the Henry's law constant. The plots illustrate both the experimentally measured data and the DFT-predicted (labeled "theory") trends.

Error analysis was conducted on the calculated equilibrium partial pressures and solubilities for the Henry's law graphs. For the partial pressures, error is derived from the uncertainty of P_{eq} (±0.08%, from transducer) and subsequent subtraction of Ar partial pressure (38.61 ± 0.83 kPa). Due to a lack of data, the vapor pressure uncertainty could not be considered. For solubility, error is primarily derived from uncertainties in the initial moles, final moles, and electrolyte volume values. Specifically, since moles values are calculated assuming ideal gas behavior [n = PV/(RT)], error in the initial moles value includes uncertainty in the transducer pressure and the gas compartment volume measurement; error in the final moles value includes uncertainty in the total volume of the cell and the partial pressure of target gas. The electrolyte volume uncertainty is given as ±0.1 mL from the syringe manufacturer.

For the gases studied in this work, the Henry's law coefficients predicted by DFT differ from the experimental values by factors of 1.43 and 2.72 for C₂H₄ and CO₂, respectively. This discrepancy implies that the error in the predicted solvation free energy is at most 0.63 kcal mol⁻¹ (0.03 eV), which is well within "chemical accuracy" (of 1 kcal mol^{-1}). In part, the observed error between theory and experiment may arise because the SMD parameters used in this study to calculate k did not account for the effect of FEC nor the $LiPF_6$ salt, which may impact the solvation free energy of small molecules. However, we attribute the error in calculated solvation free energy primarily to the fundamental limitations of implicit solvent models. As mentioned, SMD performs well on a variety of neutral small molecules in organic solvents, but it is nonetheless known to fail to capture certain effects, for instance ionic and hydrogen bonding.⁴⁵ Calculating solvation free energies using explicit solvation shells, rather than an

pubs.acs.org/jced



Figure 2. Experimental and model fit pressure decay curves of the t = 1060 h solubility trial of C₂H₄, ending at (a) 48, (b) 96, (c) 120, and (d) 144 h. Each trial produces a $P_{eq,fit}$ from eq 2 that is compared to the experimental $P_{eq,data}$ measured at 1060 h. From this data it was determined that at least 96 h is a sufficient amount of time for the fit to predict an accurate P_{eq} for C₂H₄ at T = 303.15 K.



Figure 3. Experimental and model fit pressure decay curves with R^2 comparisons for CO₂ and C₂H₄ at P_{gr} = 193 kPa and T = 303.15 K.

implicit solvent medium, may provide an opportunity to achieve better agreement with experiment.

DISCUSSION

The present work describes an experimental procedure to measure the solubility of gases in a commonly used lithium-ion battery electrolyte by monitoring the pressure change of a gas as it dissolves into the liquid electrolyte. The gases in this study were chosen because they readily form via (electro) chemical reactions during normal operation of a Li-ion battery during cell formation. While the time to reach equilibrium pressure and saturation gas concentration is typically 500–1000 h, it is shown here that a multiphase model allows extrapolation to equilibrium with measurements lasting <100 h. As the equilibrium constant for a given gas species is highly influenced by the electrolyte composition, including the salt concentration, it would be intractable to measure many different gas/ electrolyte equilibrium concentrations without the approach described in this work.



Figure 4. Experimental and theoretical Henry's law curves of (a) C_2H_4 and (b) CO_2 in GenF3 electrolyte at T = 303.15 K and $50 < P_{eq} < 128$ kPa. The Henry's law coefficients predicted by DFT differ from the experimental values by factors of 1.43 and 2.72 for C_2H_4 and CO_2 , respectively.

This work also demonstrates the calculation of Henry's law constants for these gas phase species in carbonate electrolyte, enabling the determination of their equilibrium dissolved

P _{gr} , kPa ^a	PP _{i,C₂H₄} , kPa ^b	<i>PP</i> _{eq,C2H4} , kPa ^c	$\chi_{\rm ,C_2H_4} (10^3)^d$
133.4	71.7	57.0	3.2
132.8	71.4	55.9	3.4
135.0	72.6	58.5	3.1
198.4	107	85.7	4.5
207.7	112	91.3	4.4
202.0	109	87.1	4.6
288.4	155	126.6	6.1
289.4	156	127.2	6.1
288.1	155	128.1	5.8

^{*a*}C₂H₄ pressure of gas reservoir. Relative uncertainty $u_r(P_{gr}) = 0.0009$. ^{*b*}Initial partial pressure of gas above the liquid, calculated from P_{gr} . Relative uncertainty $u_r(PP_{i,C_2H_4}) = 0.009$. ^{*c*}Final partial pressure of gas above the liquid. Standard uncertainty $u(PP_{eq,C_2H_4}) = 0.8$. ^{*d*}Solubility of gas in mole fraction. Relative standard uncertainty $u_r(\chi_{C_2H_4}) = 0.08$ for $P_{gr} = 132.8 - 135.0$, $u_r(\chi_{C_2H_4}) = 0.07$ for 298.4 < $P_{gr} < 288.4$.

Table 3. Measured Experimental Data Used to Calculate CO_2 in GenF3 Electrolyte Henry's Law Constant, $k = 1.1 \times 10^4$ kPa, $u(k_{CO_2}) = 0.04$, at T = 303.15 K

P _{gr} , kPa ^α	PP _{i,CO2} , kPa [₺]	<i>PP</i> _{eq,CO2} , kPa ^c	$\chi_{, CO_2} (10^3)^d$
130.9	70.4	49.0	4.6
130.8	70.3	50.2	4.3
202.6	109	75.1	7.3
204.5	110	76.5	7.2
289.2	155	107.7	10.3
290.8	156	108.8	10.2

^{*a*}CO₂ pressure of gas reservoir. Relative uncertainty $u_r(P_{gr}) = 0.0009$. ^{*b*}Initial partial pressure of gas above the liquid, calculated from P_{gr} . Relative uncertainty $u_r(PP_{i,CO_2}) = 0.009$. ^{*c*}Final partial pressure of gas above the liquid. Standard uncertainty $u(PP_{eq,CO_2}) = 0.8$. ^{*d*}Solubility of gas in mole fraction. Relative standard uncertainty $u_r(\chi_{,CO_2}) = 0.06$ for $P_{gr} = 130.9$, 130.8, $u_r(\chi_{,CO_2}) = 0.05$ for 202.6 < $P_{gr} < 290.8$.

concentrations. We anticipate that this will allow researchers to better interrogate the complex, competing reaction pathways, including gas reactants, occurring in Li-ion batteries, particularly within the SEI. Furthermore, there are relatively few Henry's law constants reported in the literature in batteryrelevant systems.^{28,29} Thus, there is an inherent need to measure and report the Henry's law constants for predominant gas species in common Li-ion battery electrolytes.

Additionally, the thermodynamic calculations from DFT are validated by comparison to experimentally determined Henry's law constants for the two gaseous species studied (CO₂ and C₂H₄) in this work. The DFT-predicted and measured Henry's law constants were found to agree well, with small errors ≤ 0.63 kcal mol⁻¹. This indicates that current theoretical models can predict Henry's law constants within an order of magnitude, potentially even without accounting for every species in solution. However, further improvements can still be made, and there is a need to develop physics-based models that can accurately determine these constants to extrapolate the present results to additional electrolyte systems.

CONCLUSIONS

The pressure decay at different starting pressures of C₂H₄ and CO₂ gas dissolving into GenF3 battery electrolyte was recorded and modeled to extrapolate the equilibrium pressures, P_{eq} . Analysis of trial run time was conducted, determining the minimum trial length of 96 h to minimize percent error of the model fit P_{eq} . The differences in moles from initial to equilibrium partial pressures were used to calculate the solubility of each gas at the equilibrium pressures, and this was done for several initial pressures. The equilibrium pressures versus solubility in mole fraction were plotted to calculate the Henry's law constants k for each gas with $k_{C_2H_4} = 2.0 \times 10^4$ kPa and $k_{CO_2} = 1.1 \times 10^4$ kPa. Using the Environmental Protection Agency (EPA) standards, these two species would be considered "volatile" in the electrolyte studied.^{46,47} This means that the species are significantly more stable in the gas-phase as compared to the liquid-phase. In terms of studying reaction mechanisms that form SEI species, there is a significant competing pathway to eject these dissolved species to the gas-phase compared to retaining these species to feed additional reaction cascades. Nevertheless, the equilibrium saturation concentrations of both C₂H₄ and CO₂ are estimated to be in the range of 0.5-1.0 mol % at reasonable partial pressures expected for a typical Li-ion battery, suggesting that an appreciable amount of gas remains in the liquid that could contribute to electrochemical side reactions during cell operation.

These experimental constants were compared to theoretical k constants in 3:7 wt/wt EC/EMC solution. Despite the DFT calculations not including salt coordination or FEC effects, the predicted constants differed by factors of 1.43 and 2.72 for C_2H_4 and CO_2 , respectively. Differences between experimental measurements and DFT calculations are attributed to limitations of the DFT solvent models and presence of additives in the EC/EMC experimental solution.

Although only two gases were measured in this study, our approach will be extended to several others—CO, C_2H_2 , H_2 , CH_4 , and O_2 —that are typically formed during formation and cycling of a lithium-ion battery. Furthermore, additional electrolyte formulations will be explored to quantify effects on gas solubility of varying solvent, liquid additives, and salt concentrations.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.jced.3c00692.

Supporting Information that includes an annotated photo of the pressure decay apparatus used in this study, Figure S1; example of a pressure decay measurement of carbon dioxide gas in water, Figure S2, using the apparatus described in this work; and measurement of the Henry's law constant for CO_2 in water at T = 303 K, Figure S3 (PDF)

AUTHOR INFORMATION

Corresponding Author

Bertrand J. Tremolet de Villers – National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States; occid.org/0000-0001-8685-539X; Email: bertrand.tremolet@nrel.gov

Authors

- Mel Soto National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States
- Kae Fink National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States; Ocid.org/0000-0001-9363-4632
- Christof Zweifel National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States
- Peter J. Weddle National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States; orcid.org/0000-0002-1600-0756
- Evan Walter Clark Spotte-Smith Department of Materials Science and Engineering, University of California, Berkeley, California 94720, United States; Materials Science Division, Lawerence Berkeley National Laboratory, Berkeley, California 94720, United States; © orcid.org/0000-0003-1554-197X
- Gabriel M. Veith Chemical Sciences Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, United States; o orcid.org/0000-0002-5186-4461
- Kristin A. Persson Department of Materials Science and Engineering, University of California, Berkeley, California 94720, United States; Materials Science Division and Molecular Foundry, Lawerence Berkeley National Laboratory, Berkeley, California 94720, United States
- Andrew M. Colclasure National Renewable Energy Laboratory (NREL), Golden, Colorado 80401, United States; © orcid.org/0000-0002-9574-5106

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.jced.3c00692

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

This work is authored in part by the National Renewable Energy Laboratory (M.S., K.F., C.Z., P.J.W., A.M.C. and B.T.d.V.), operated by Alliance for Sustainable Energy, LLC, for the U.S. Department of Energy (DOE) under Contract no. DE-AC36-08GO₂8308. This research was supported by the U.S. Department of Energy's Vehicle Technologies Office under the Silicon Consortium Project, directed by Brian Cunningham and managed by Anthony Burrell. This work was also supported in part by the U.S. Department of Energy, Office of Science, Office of Workforce Development for Teachers and Scientists (WDTS) under the Science Undergraduate Laboratory Internships Program (SULI) (M.S. and C.Z.). Additional support was provided by the Kavli Energy NanoScience Institute Philomathia Graduate Student Fellowship (E.W.C.S.-S.). Data for this study was produced using computational resources provided by the National Energy Research Scientific Computing Center (NERSC), a U.S. Department of Energy Office of Science User Facility under Contract no. DE-AC02-05CH11231, the Eagle HPC system at the National Renewable Energy Laboratory (NREL), and the Lawrencium HPC cluster at Lawrence Berkeley National Laboratory (P.J.W., E.W.C.S.-S., A.M.C., and K.A.P.). A portion of this manuscript has been authored by UT-Battelle, LLC (G.M.V.), under Contract DE-AC05-00OR22725 with the U.S. Department of Energy. The views expressed in the article do not necessarily represent the views of the DOE or the U.S. Government. The U.S. Government retains and the

publisher, by accepting the article for publication, acknowledges that the U.S. Government retains a nonexclusive, paidup, irrevocable, worldwide license to publish or reproduce the published form of this work, or allow others to do so, for U.S. Government purposes.

REFERENCES

(1) Armand, M.; Tarascon, J. M. Building better batteries. *Nature* **2008**, 451, 652–657.

(2) Placke, T.; Kloepsch, R.; Dühnen, S.; Winter, M. Lithium ion, lithium metal, and alternative rechargeable battery technologies: the odyssey for high energy density. *J. Solid State Electrochem.* **2017**, *21*, 1939–1964.

(3) Dunn, B.; Kamath, H.; Tarascon, J.-M. Electrical Energy Storage for the Grid: A Battery of Choices. *Science* **2011**, *334*, 928–935.

(4) Eshetu, G.; Zhang, H.; Judez, X.; Adenusi, H.; Armand, M.; Passerini, S.; Figgemeier, E. Production of high-energy Li-ion batteries comprising silicon-containing anodes and insertion-type cathodes. *Nat. Commun.* **2021**, *12*, 5459.

(5) Eshetu, G. G.; Figgemeier, E. Confronting the Challenges of Next-Generation Silicon Anode-Based Lithium-Ion Batteries: Role of Designer Electrolyte Additives and Polymeric Binders. *ChemSusChem* **2019**, *12*, 2515–2539.

(6) McBrayer, J.; Rodrigues, M. T. F.; Schulze, M. C.; Abraham, D. P.; Apblett, C. A.; Bloom, I.; Carroll, G. M.; Colclasure, A. M.; Fang, C.; Harrison, K. L.; et al. Calendar aging of silicon-containing batteries. *Nat. Energy* **2021**, *6*, 866–872.

(7) Schulze, M.; Rodrigues, M. T. F.; McBrayer, J. D.; Abraham, D. P.; Apblett, C. A.; Bloom, I.; Chen, Z.; Colclasure, A. M.; Dunlop, A. R.; Fang, C.; et al. Critical Evaluation of Potentiostatic Holds as Accelerated Predictors of Capacity Fade during Calendar Aging. *J. Electrochem. Soc.* **2022**, *169*, 050531.

(8) Liu, B.; Zhang, J.-G.; Xu, W. Advancing Lithium Metal Batteries. *Joule* **2018**, *2*, 833–845.

(9) Liu, J.; Bao, Z.; Cui, Y.; Dufek, E. J.; Goodenough, J. B.; Khalifah, P.; Li, Q.; Liaw, B. Y.; Liu, P.; Manthiram, A.; et al. Pathways for practical high-energy long-cycling lithium metal batteries. *Nat. Energy* **2019**, *4*, 180–186.

(10) Sun, Y.-K. High-Capacity Layered Cathodes for Next-Generation Electric Vehicles. ACS Energy Lett. 2019, 4, 1042–1044. (11) Gutierrez, A.; Tewari, D.; Chen, J.; Srinivasan, V.; Balasubramanian, M.; Croy, J. Review—Earth-Abundant, Mn-Rich Cathodes for Vehicle Applications and Beyond: Overview of Critical Barriers. J. Electrochem. Soc. 2023, 170, 030509.

(12) Cao, X.; Jia, H.; Xu, W.; Zhang, J.-G. Review—Localized High-Concentration Electrolytes for Lithium Batteries. *J. Electrochem. Soc.* **2021**, *168*, 010522.

(13) Bläubaum, L.; Röse, P.; Schmidt, L.; Krewer, U. The Effects of Gas Saturation of Electrolytes on the Performance and Durability of Lithium-Ion Batteries. *ChemSusChem* **2021**, *14*, 2943–2951.

(14) Ellis, L. D.; Allen, J. P.; Thompson, L. M.; Harlow, J. E.; Stone, W. J.; Hill, I. G.; Dahn, J. R. Quantifying, Understanding and Evaluating the Effects of Gas Consumption in Lithium-Ion Cells. *J. Electrochem. Soc.* **2017**, *164*, A3518.

(15) Plichta, E.; Slane, S.; Uchiyama, M.; Salomon, M.; Chua, D.; Ebner, W.; Lin, H. An Improved $\text{Li}/\text{Li}_x\text{CoO}_2$ Rechargeable Cell. *J. Electrochem. Soc.* **1989**, *136*, 1865–1869.

(16) Youngman Chusid, O.; Ein Ely, E.; Aurbach, D.; Babai, M.; Carmeli, Y. Electrochemical and spectroscopic studies of carbon electrodes in lithium battery electrolyte systems. *J. Power Sources* **1993**, 43, 47–64.

(17) Hopkins, E.; Frisco, S.; Pekarek, R.; Stetson, C.; Huey, Z.; Harvey, S.; Li, X.; Key, B.; Fang, C.; Liu, G.; Yang, G.; Teeter, G.; Neale, N.; Veith, G. Examining CO_2 as an Additive for Solid Electrolyte Interphase Formation on Silicon Anodes. *J. Electrochem. Soc.* **2021**, *168*, 030534.

(18) Krause, L. J.; Chevrier, V. L.; Jensen, L. D.; Brandt, T. The Effect of Carbon Dioxide on the Cycle Life and Electrolyte Stability of

Li-Ion Full Cells Containing Silicon Alloy. J. Electrochem. Soc. 2017, 164, A2527–A2533.

(19) Aurbach, D.; Ein-Eli, Y.; Chusid Youngman, O.; Carmeli, Y.; Babai, M.; Yamin, H. The Correlation Between the Surface Chemistry and the Performance of Li-Carbon Intercalation Anodes for Rechargeable 'Rocking-Chair' Type Batteries. *J. Electrochem. Soc.* **1994**, *141*, 603–611.

(20) Ein-Eli, Y.; Markovsky, B.; Aurbach, D.; Carmeli, Y.; Yamin, H.; Luski, S. The dependence of the performance of Li-C intercalation anodes for Li-ion secondary batteries on the electrolyte solution composition. *Electrochim. Acta* **1994**, *39*, 2559–2569.

(21) Strehle, B.; Solchenbach, S.; Metzger, M.; Schwenke, K. U.; Gasteiger, H. A. The Effect of CO 2on Alkyl Carbonate Trans-Esterification during Formation of Graphite Electrodes in Li-Ion Batteries. *J. Electrochem. Soc.* **2017**, *164*, A2513–A2526.

(22) Hou, T.; Yang, G.; Rajput, N. N.; Self, J.; Park, S.-W.; Nanda, J.; Persson, K. A. The influence of FEC on the solvation structure and reduction reaction of LiPF6/EC electrolytes and its implication for solid electrolyte interphase formation. *Nano Energy* **2019**, *64*, 103881.

(23) Weddle, P. J.; Spotte-Smith, E.; Verma, A.; Patel, H.; Fink, K.; Tremolet de Villers, B. J.; Schulze, M.; Blau, S.; Smith, K.; Persson, K.; Colclasure, A. Continuum-level modeling of Li-ion battery SEI by upscaling atomistically informed reaction mechanisms. *Electrochim. Acta* **2023**, *468*, 143121.

(24) Seitzinger, C. L.; Sacci, R. L.; Coyle, J. E.; Apblett, C. A.; Hays, K. A.; Armstrong, R. R.; Rogers, A. M.; Armstrong, B. L.; Bennet, T. H.; Neale, N. R.; Veith, G. M. Intrinsic Chemical Reactivity of Silicon Electrode Materials: Gas Evolution. *Chem. Mater.* **2020**, *32*, 3199–3210.

(25) Fang, C.; Tran, T.-N.; Zhao, Y.; Liu, G. Electrolyte decomposition and solid electrolyte interphase revealed by mass spectrometry. *Electrochim. Acta* **2021**, *399*, 139362.

(26) Spotte-Smith, E.; Blau, S.; Xie, X.; Patel, H.; Wen, M.; Wood, B.; Dwaraknath, S.; Persson, K. Quantum chemical calculations of lithium-ion battery electrolyte and interphase species. *Sci. Data* **2021**, *8*, 203.

(27) Chevrier, V.; Krause, L.; Jensen, L.; Huynh, C.; Triemert, M.; Bowen, E.; Thorson, J. Design of Positive Electrodes for Li-Ion Full Cells with Silicon. *J. Electrochem. Soc.* **2018**, *165*, A2968–A2977.

(28) de la Iglesia, O.; Mainar, A.; Pardo, J. I.; Urieta, J. Solubilities of Nonpolar Gases in Triethylene Glycol Dimethyl Ether, Tetraethylene Glycol Dimethyl Ether, Dimethyl Carbonate, and Diethyl Carbonate at 298.15 K and 101.33 kPa Partial Pressure of Gas. *J. Chem. Eng. Data* 2003, 48, 657–661.

(29) Terrado, E.; Pardo, J.; Urieta, J.; Mainar, A. Solubilities of Nonpolar Gases in Dimethyl Carbonate and Diethyl Carbonate. *J. Chem. Eng. Data* 2005, *50*, 512–516.

(30) Dougassa, Y.; Jacquemin, J.; El Ouatani, L.; Tessier, C.; Anouti, M. Viscosity and carbon dioxide solubility for LiPF_{6} , LiTFSI, and LiFAP in alkyl carbonates: Lithium salt nature and concentration effect. J. Phys. Chem. B **2014**, 118, 3973–3980.

(31) Dougassa, Y.; Jacquemin, J.; El Ouatani, L.; Tessier, C.; Anouti, M. Low pressure methane solubility in lithium-ion batteries based solvents and electrolytes as a function of temperature. Measurement and prediction. *J. Chem. Thermodyn.* **2014**, *79*, 49–60.

(32) Behzadfar, E.; Hatzikiriakos, S. Diffusivity of CO_2 in Bitumen: Pressure–Decay Measurements Coupled with Rheometry. *Energy Fuels* **2014**, *28*, 1304–1311.

(33) Zhang, Y. P.; Hyndman, C. L.; Maini, B. B. Measurement of gas diffusivity in heavy oils. *J. Pet. Sci. Eng.* **2000**, *25*, 37–47.

(34) Browning, K. L.; Baggetto, L.; Unocic, R. R.; Dudney, N. J.; Veith, G. M. Gas evolution from cathode materials: A pathway to solvent decomposition concomitant to SEI formation. *J. Power Sources* **2013**, 239, 341–346.

(35) Hemmerling, J.; Schäfer, J.; Jung, T.; Kreher, T.; Ströbel, M.; Gassmann, C.; Günther, J.; Fill, A.; Birke, K. P. Investigation of internal gas pressure and internal temperature of cylindrical Li-ion cells to study thermodynamical and mechanical properties of hard case battery cells. *J. Energy Storage* **2023**, *59*, 106444.

(36) IUPAC-NIST Solubility Database, Version 1.1-NIST Standard Reference Database 106, 2012, https://srdata.nist.gov/solubility/.

(37) Hess, S.; Wohlfahrt-Mehrens, M.; Wachtler, M. Flammability of Li-Ion Battery Electrolytes: Flash Point and Self-Extinguishing Time Measurements. *J. Electrochem. Soc.* **2015**, *162*, A3084–A3097.

(38) Epifanovsky, E.; Gilbert, A. T. B.; Feng, X.; Lee, J.; Mao, Y.; Mardirossian, N.; Pokhilko, P.; White, A. F.; Coons, M. P.; Dempwolff, A. L.; et al. Software for the frontiers of quantum chemistry: An overview of developments in the Q-Chem 5 package. *J. Chem. Phys.* **2021**, *155*, 084801.

(39) Mardirossian, N.; Head-Gordon, M. ω B97X-V: A 10parameter, range-separated hybrid, generalized gradient approximation density functional with nonlocal correlation, designed by a survival-of-the-fittest strategy. *Phys. Chem. Chem. Phys.* **2014**, *16*, 9904–9924.

(40) Rappoport, D.; Furche, F. Property-optimized Gaussian basis sets for molecular response calculations. *J. Chem. Phys.* **2010**, *133*, 134105.

(41) Marenich, A.; Cramer, C.; Truhlar, D. Universal Solvation Model Based on Solute Electron Density and on a Continuum Model of the Solvent Defined by the Bulk Dielectric Constant and Atomic Surface Tensions. J. Phys. Chem. B **2009**, 113, 6378–6396.

(42) Mennucci, B. Polarizable continuum model. Wiley Interdiscip. Rev. Comput. Mol. Sci. 2012, 2, 386–404.

(43) Modarresi, H.; Modarress, H.; Dearden, J. C. Henry's law constant of hydrocarbons in air-water system: The cavity ovality effect on the non-electrostatic contribution term of solvation free energy. SAR QSAR Environ. Res. 2005, 16, 461–482.

(44) Schüürmann, G. Prediction of Henry's law constant of benzene derivatives using quantum chemical continuum-solvation models. *J. Comput. Chem.* **2000**, *21*, 17–34.

(45) Zhang, J.; Zhang, H.; Wu, T.; Wang, Q.; van der Spoel, D. Comparison of implicit and explicit solvent models for the calculation of solvation free energy in organic solvents. *J. Chem. Theory Comput.* **2017**, *13*, 1034–1043.

(46) McAtee, M.; Johnson, J.; Hauschild, V.; et al. Methodology for developing chemical exposure guidelines for deployed military personnel; U.S. Army Public Health Command, 2010, Available at https://phcameddarmymil/PHCResourceLibrary/ RD230June2010Revision.pdf.

(47) Murphy, B.; Morrison, R. Introduction to Environmental Forensics; Elsevier, 2007.